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Polarized recombination of acoustically transported carriers in GaAs nanowires

Michael Möller^{1*}, Alberto Hernández-Mínguez², Steffen Breuer², Carsten Pfüller², Oliver Brandt², Mauricio M de Lima Jr^{1,3}, Andrés Cantarero¹, Lutz Geelhaar², Henning Riechert² and Paulo V Santos²

Abstract

The oscillating piezoelectric field of a surface acoustic wave (SAW) is employed to transport photoexcited electrons and holes in GaAs nanowires deposited on a SAW delay line on a LiNbO₃ crystal. The carriers generated in the nanowire by a focused light spot are acoustically transferred to a second location where they recombine. We show that the recombination of the transported carriers occurs in a zinc blende section on top of the predominant wurtzite nanowire. This allows contactless control of the linear polarized emission by SAWs which is governed by the crystal structure. Additional polarization-resolved photoluminescence measurements were performed to investigate spin conservation during transport.

Keywords: Charge transport, Spin transport, GaAs, Nanowires, Surface acoustic waves, Photoluminescence, Polarization

Background

Semiconductor nanowires offer new perspectives for lowdimensional semiconductor devices since the small radius favors mesoscopic size effects and lifts the epitaxial constraints associated with the growth of dissimilar materials. In addition, the geometry of nanowires enables them to function as active device elements and interconnects, which can lead to highly integrated optoelectronic device structures. In this respect, the crystal structure of the nanowires is of importance since it controls the optoelectronic properties. Exploitation of these properties normally requires doping and contacting of non-planar, nanometer-sized structures for the application of electric control fields. Here, we applied surface acoustic waves (SAW) to remotely control the carrier recombination in GaAs/AlGaAs core-shell nanowires.

We have recently demonstrated that the oscillating piezoelectric field of a SAW can transport photoexcited charge carriers in GaAs nanowires as well as control the spatial location of exciton recombination along the nanowire axis [1]. In this contribution, we use polarized photoluminescence (PL) spectroscopy with spatial

*Correspondence: michael.moeller@uv.es



Methods

The nanowires used in our study were grown by molecular beam epitaxy (MBE) using a self-assisted vapor-liquidsolid growth process on Si(111) substrate [2]. They consist of an undoped GaAs core of 106 ± 18 nm diameter coated with $22 \pm 9 \text{ nm Al}_x \text{Ga}_{1-x} \text{As shell with } x = 0.1$. The nanowires, which have a predominant wurtzite structure, were then mechanically dispersed on the surface of a 128°-rotated Y-cut X-propagating LiNbO3 crystal containing two floating electrode unidirectional transducers designed to generate SAWs at multiple harmonics of the fundamental wavelength $\lambda_{SAW} = 35 \,\mu$ m. The experiments were carried out using the second harmonic with $\lambda_{\text{SAW}} = 17.5 \,\mu\text{m}$, which correspond to the SAW frequency and period of $f_{\text{SAW}} = 226.5 \text{ MHz}$ and $\tau_{\text{SAW}} =$ 4.4 ns, respectively. The experiments were performed on those single nanowires which were incidentally aligned with the SAW propagation direction (see Figure 1a).

The results shown here were obtained at low temperatures (20 K) by mounting the sample in a microscope



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¹ Materials Science Institute, University of Valencia, Catedrático José Beltrán 2, Paterna 46980, Spain

Full list of author information is available at the end of the article



cryostat for spatially resolved photoluminescence measurements (see Figure 1a). A 757-nm diode laser yielding 150 ps wide pulses was used as excitation source, the repetition rate of the laser being synchronized with the SAW frequency. The linearly polarized laser beam was focused onto the nanowire by using a $50 \times$ objective (spot diameter of approximately 1.5 μ m on the sample). The PL emission of the nanowire, collected by the same objective, was split by a birefringence prism into two orthogonally polarized rays and imaged on the entrance slit of a spectrometer. The rays with polarization parallel and perpendicular to the nanowire axis are then detected with spatial resolution (about $1 \mu m$) on the upper and lower regions of a cooled charge-coupled device camera (CCD), respectively. For the spin transport measurements, a $\lambda/4$ plate was put into the beam path before the objective in order to convert the linear polarized light into a left-handed circularly polarized one and vice versa (see Figure 1b).

Results and discussion

Figure 2a,b displays the PL images excited by a tightly focused laser spot close to the end of the nanowire facing the acoustic transducer (cf. diagrams on the left of Figure 2) in the absence (Figure 2a) and presence (Figure 2b) of a SAW. In the upper and lower part of the images, the spectrally (horizontal scale) and spatially (vertical scale) resolved nanowire emission polarized parallel and perpendicular to the nanowire axis are represented respectively. The corresponding spatial PL profiles integrated from 796.6 to 823.3 nm are depicted in Figure 2c,d. Without acoustic power (Figure 2a), the emission is restricted to the region close to the excitation spot (position G in the diagrams on the left of Figure 2) and is enlarged due to diffusion processes along the nanowire. When a SAW is applied, the PL intensity at G reduces and a second emission at a remote point (position R) appears a few micrometers away from G along the SAW propagation direction. The reduction at G is due to the induced spatial separation of electrons and holes by the SAW piezoelectric field, which decreases the radiative recombination probability at G. The emission at R is attributed to the transport of the photoexcited electrons and holes along the nanowire axis and their recombination at the remote position, as we have recently demonstrated [1]. The slight PL emission appearing between G and R are associated to the recombination of some carriers at traps as stacking faults, impurities, or defects.

Without acoustic power, the broad emission band at G is highly polarized perpendicular to the nanowire axis (see Figure 2a,c). When a SAW is applied, the emission at G maintains its polarization behavior while the recombination at R is polarized parallel to the nanowire axis (see Figure 2b,d). The emission at G is consistent with the optical selection rules expected for the wurtzite GaAs nanowires [3] where the band edge emission is dipole allowed only if the electric dipole moment is perpendicular to the wurtzite *c*-axis [4]. The emission at R, however, does not coincide with these selection rules. Since the recombination of the transported charge carriers take place at a remote position the crystal structure of that



region may be different. Indeed, the emission energy of approximately 818 nm - which is exactly the band gap energy of zinc blende bulk GaAs [5] - suggests that recombination takes place in a zinc blende region at the top of the nanowire, opposite to the acoustic transducer. This is supported by transmission electron microscopy measurements, which reveal a zinc blende section in the top region of the nanowires probably created during the cooling down after growth, as previously observed in CBEand MBE-grown GaAs nanowires [6,7]. The emission polarized parallel to the nanowire axis at R is consistent with the emission characteristics observed in zinc blende GaAs nanowires [8,9]. Here, the emission is preferentially polarized along the nanowire axis due to the dielectric mismatch between the nanowire and its surroundings. The degree of linear polarization, $P = (I_{\perp} - I_{\parallel})/(I_{\perp} + I_{\parallel})$ where I_{\perp} and I_{\parallel} are the PL polarization intensities detected perpendicular and parallel to the nanowire axis, respectively - at the excitation spot G, is approximately

+88% (perpendicular) in the absence of a SAW, whereas at the remote position R the polarization becomes equal to approximately -70% (parallel) with acoustic power.

Similar polarization results have been obtained for all probed nanowires, independent of the polarization of the exciting laser. This also implies that the electron spins are not conserved during transport. To further support the last conclusion, we have carried out transport experiments where spin-up and spin-down electrons were generated on one extreme of the nanowire using right-handed (σ^+) and left-handed (σ^-) circularly polarized light, respectively. The circular polarization of the PL was detected with spatial resolution using $\lambda/4$ plate and a birefringence prism (see Figure 1). For simplicity, only the right-handed, circularly polarized emission (I_+) is considered in the following discussion. For incident right-handed, circularly polarized light (σ^+), the spatially resolved I_+ profiles in the absence and presence of a SAW are shown in Figure 3a,b, respectively. Figure 3c,d



presents the I_+ profiles along the nanowire axis integrated around the spectral positions 811 nm (blue) and 818 nm (red), in the absence and presence of a SAW for σ^+ (lines) and σ^- (circles). Without acoustic power, the PL is emitted around 811 nm at the excitation spot G (Figure 3a). When a SAW is applied, a remote spot appears emitting at 818 nm (Figure 3b), as observed in the linear polarization measurements. From Figure 3c,d one sees that I_+ is the same for σ^+ and σ^- excitations (cf. solid lines with open circles) at G. This implies that the initially generated spin-up or spin-down electrons have lost their spin polarization in the wurtzite phase at G leading to evenly distributed spin states. The same result is observed for the recombination along the wire and at R where no difference of I_+ of the transported electrons between σ^+ and σ^- can be observed.

Conclusions

We have investigated the acoustic transport of charge along GaAs nanowires. We have shown that the transported electrons and holes recombine in zinc blende sections within the predominant wurtzite nanowire. In addition, spin polarization-resolved photoluminescence

measurements have revealed that the spin polarization is lost during transport.

Competing interests

The authors declare that they have no competing interests.

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Author details

¹Materials Science Institute, University of Valencia, Catedrático José Beltrán 2, Paterna 46980, Spain. ²Paul-Drude-Institut für Festkörperelektronik, Hausvogteiplatz 5-7, Berlin 10117, Germany. ³Fundació General de la Universitat de València, Valencia 46010, Spain.

Authors' contributions

MM, AHM, and PVS performed the experiments on acoustic transport and analyzed the data. SB grew the NW sample and CP developed a process for harvesting NWs from as-grown ensembles and for dispersing them onto the LiNbO₃ substrates. LG supervised NW growth and, together with OB, AC and HR, contributed to the analyses of the data and discussions. PVS conceived the experiments, supervised the project and, together with MM and MMLJ, prepared the manuscript. All authors read and approved the final manuscript.

Authors' information

MM is PhD student, MMLJ is senior researcher and AC is the director of Materials Science Institute of the University of Valencia. AHM, SB and CP are post-doc researchers. LG, OB and PVS are senior researchers and HR is the director of the Paul-Drude-Institut für Festkörperelektronik, Berlin.

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